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(54) Title: INTERPENETRATING POLYMERIC NETWORKS COMPRISING POLYTETRAFLUOROETHYLENE AND POLYSILOXANE			
(57) Abstract <p>A composition having interpenetrating matrices comprising (a) a first polymer network characterized by nodes interconnected by fibrils and (b) a second polymer network comprising diorganosiloxy units. In another aspect, there are provided compositions comprising an interpenetrating polymer network of polytetrafluoroethylene and polydiorganosiloxane. Methods for making the foregoing compositions are also provided. <u>Strong porous films and cable insulation may be prepared from the compositions of the invention.</u></p> <p><i>Keep in file OK for name</i></p> <p><i>no - don't extract polymer</i></p>			

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"INTERPENETRATING POLYMERIC NETWORKS COMPRISING
POLYTETRAFLUOROETHYLENE AND POLYSILOXANE"

5

Background of the Invention

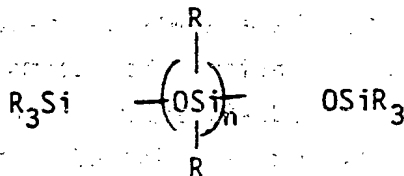
10 The present invention relates to polymeric structures having interpenetrating matrices in cured form, comprising a polytetrafluoroethylene (PTFE) network and a polysiloxane network. More particularly, the present invention relates to interpenetrating polymeric networks wherein a curable silicone composition is mixed with PTFE powder and, thereafter, said silicone is cured while simultaneously stretching the silicone/PTFE mixture.

15 Tetrafluoroethylene polymers and, in particular, polytetrafluoroethylene, are gaining more and more uses because of their chemical inertness and desirable physical properties such as water repellency and electrical insulating abilities. In one very large area, the field of porous articles, their use has been substantially blocked by the very considerable
20 difficulty of making an article porous and keeping it so while providing it with adequate strength. Complicated and expensive processes have been devised such as adding a filler to the polymer prior to shaping and then removing the filler after shaping, for example, by leaching it out of the shaped article
25 with a solvent, or by melting or burning it out.

Gore, U.S. Patent Nos. 3,953,566; 3,962,153; 4,096,227; and 4,187,390, overcame the difficulties of the prior art by stretching PTFE rods as fast as possible so as to provide

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tetrafluoroethylene polymer having outstanding combinations of high porosity and high strength. Moreover, even though the rod was greatly lengthened its diameter remained virtually unchanged throughout its length. The rapid stretching also transformed the hard, shiny rods into a soft, flexible material. Gore also teaches that there may be included as fillers, silicone fluids having structures represented by the formula



where n is from 0 to about 2000 and R is an alkyl or aryl radical.

Products based on rapidly stretching highly crystalline polytetrafluoroethylene include porous films for filters and laminates; fabric laminates of PTFE film bonded to fabric to produce a material having the contradictory properties of impermeability to liquid water and permeability to water vapor, the material being used to make "breathable" rainwear and filters; porous yarn for weaving and braiding into other products such as pump packing; tubes used as replacements for human arteries and veins; and insulation for high performance electric cables.

An interpenetrating polymer network (IPN) is a combination of at least two polymers in network form, at least one of which is synthesized and/or crosslinked in the immediate presence of the other. IPN's can be further classified as sequential IPN's, where the networks are made sequentially, and simultane-

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ous IPN's, where the monomers are mixed together and polymerized by independent and non-interfering routes. Sequential IPN's can also be prepared by swelling a polymer with a monomer and activating agents and, thereafter, polymerizing the monomer in situ.

Since true IPN's require mixing on the molecular level, which is not possible with some polymers due to their well known thermodynamic incompatibility, those skilled in the art often refer to these IPN's as pseudo-IPN's. For purposes of the present invention the terms "interpenetrating polymer network" and "pseudo-interpenetrating polymer network" are used interchangeably and are intended to describe both sequential IPN's and simultaneous IPN's.

Interpenetrating polymeric networks wherein one of the polymers is a polysiloxane are known in the art. Foscante et al., U.S. Pat. No. 4,250,074, discloses an interpenetrating polymer network of a polymerized epoxy resin network intertwined with a polysiloxane network formed by the hydrolytic polycondensation of silane groups. This is achieved by the simultaneous polymerization, at substantially balanced reaction rates, of a mixture of epoxy resin and silane. An amine curing agent forms the epoxy network and water distributed throughout the mixture causes the hydrolytic polycondensation of the silane groups. A preferred method for preparing the IPN is to react epoxy resin with an aminosilane capable of both hydrolytic polycondensation of the silane moiety and amine addition of the oxirane rings of the epoxy resin.

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5 Frisch et al., U.S. Pat. No. 4,302,553, describes interpenetrating polymeric networks characterized by a single glass transition temperature, comprising at least two separate chemically dissimilar macrocyclic structures of crosslinked polymer chains which do not contain ionizable groups and which are crosslinked by different crosslinking agents which crosslink by different crosslinking mechanisms, the polymer chains of one macrocyclic structure being threaded through, but having substantially no intermolecular chemical bonding with, 10 another macrocyclic structure of the interpenetrating polymer network, with the proviso that the macrocyclic structures have rings of at least 20 ring atoms. Suitable crosslinking polymers typically can be polyfunctional polyurethane, polyepoxides, polyesters, polyamides, polyimides, phenolics, 15 polysiloxanes, polysulfides, acrylics, poly (styrenebutadiene), poly (styrene-acrylonitrile), poly (butadiene-acrylonitrile), polychloroprene, poly (chloroprene-acrylonitrile) and poly (acrylonitrile-butadiene-styrene).

20 Pierce et al., U.S. Pat. No. 4,312,920, provides a blood contacting layer and a blood contacting interface consisting of a solvent cast polyurethane alloyed with a filler-free silicone rubber. The alloy interface comprises an interpenetrating polymer network consisting of polyurethane and filler-free silicone rubber at the molecular level.

25 Siegfried et al., U.S. Pat. No. 4,468,499, relates to interpenetrating polymer networks comprised of at least two polymers, the second of which is polymerized in the presence of the first of said polymers, in a mutual solution of said first polymer and a prepolymer or monomer of said second polymer, at

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least one of said polymers comprising a continuous phase within said composition, said polymers being crosslinked by a non-covalent bonding mechanism selected from the group consisting of (a) ionic attraction and (b) block copolymers, each having at least three blocks, at least one of said blocks comprising an elastomeric block and at least two of said blocks comprising physically crosslinked glassy blocks, crystalline blocks or hydrogen bonded blocks, wherein one of said polymers is a triblock copolymer containing monoalkenyl arene polymer end blocks and the other of said polymers is a monoalkenyl arene monomer copolymerized with an acid forming monomer. The crystalline block can be polydiphenylsiloxane and the elastomeric block can be polydimethylsiloxane.

Arkles, U.S. Pat. No. 4,500,688, discloses melt processable compositions comprising a silicone component which is vulcanizable by the reaction of a hydride-containing silicone within a polymeric thermoplastic matrix to form a silicone pseudo-interpenetrating polymer network, said vulcanization of said silicone component being initiated during thermoplastic melt-mixing of said component with said matrix. The polymeric thermoplastic matrices include polyamides, thermoplastic polyurethanes, bisphenol A polycarbonates, styrenics and polyacetals.

All of the foregoing patents are incorporated by reference into the present disclosure.

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Summary of the Invention

It is an object of the present invention to provide interpenetrating polymer networks comprising a polydiorganosiloxane network and a polytetrafluoroethylene network.

5 It is another object of the present invention to provide porous materials having improved impermeability to liquid water and improved air permeability, e.g., breathability.

10 Another object of the present invention is to provide materials useful in the manufacture of apparel fabric laminates, geotextiles, geomembranes, filters, and other industrial, medical and electrical applications.

Still another object of the present invention is to provide methods for making polysiloxane - polytetrafluoroethylene interpenetrating networks.

15 In accordance with the present invention there are provided novel compositions comprising a polysiloxane component vulcanized within a polytetrafluoroethylene component to form an interpenetrating polymer network.

20 There are also provided methods of producing polysiloxane - polytetrafluoroethylene interpenetrating polymer networks by curing a silicone within the polytetrafluoroethylene matrix, particularly at elevated temperatures while simultaneously stretching a mixture of said silicone and polytetrafluoroethylene.

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Description of the Invention

In accordance with one aspect of the present invention there are provided compositions having interpenetrating matrices in cured form, comprising:

(a) a first polymer network characterized by nodes and fibrils interconnected by fibrils, and

(b) a second polymer network comprising diorganosiloxy units.

Component (a) can be any polymer capable of being stretched, drawn or expanded so as to obtain a microstructure characterized by nodes interconnected by very small fibrils. It is especially preferred that component (a) be polytetrafluoroethylene as taught by Gore in the aforementioned U.S. patents. Polyethylene, polyamides, and polyesters are also known to exhibit a fibrillar structure upon being drawn or expanded. The suitability of other polymers for practicing the present invention will be obvious to those skilled in the art or can be ascertained without undue experimentation.

Component (b) can be any curable silicone composition, however; it is preferred that an addition curable silicone composition be employed in the practice of the invention. Generally stated, addition curable silicone compositions comprise (1) a polydiorganosiloxane having alkenyl unsaturation, (2) an organohydrogenpolysiloxane crosslinking agent, and (3) a catalyst for promoting crosslinking of (1) and (2). The principles of this chemistry are well known to those skilled in the art.

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Alkenyl-containing polydiorganosiloxanes typically employed in the practice of the present invention can have viscosities up to 100,000,000 centipoise or more at 25°C, for example, in accordance with the teaching of U.S. Pat. No. 4,061,609 to Bobear. It has been found that excellent results are obtained when the viscosity of the alkenyl-containing polysiloxane is from about 500 centipoise to 50,000 centipoise at 25°C, and especially when the viscosity is from about 3000 centipoise to 6000 centipoise at 25°C.

Organohydrogenpolysiloxanes that can be utilized in the present invention may be linear or resinous and have viscosities of between about 25 centipoise and 10,000 centipoise at 25°C, with the preferred range being from about 100 centipoise to about 1000 centipoise at 25°C.

The curing catalyst can be either an organic peroxide or a precious metal containing material. Suitable organic peroxides include dibenzoyl peroxide, bis-2,4-dichlorobenzoyl peroxide, di-t-butyl peroxide, 2,5-dimethyl-2,5-di- (t-butylperoxy) hexane, and dicumyl peroxide. Precious metal containing catalysts can be based on the metals rhodium, ruthenium, palladium, osmium, irridium and platinum. It is particularly preferred that a platinum metal complex be employed as the catalyst, for example, as taught by Ashby in U.S. Pat. Nos. 3,159,601 and 3,159,662, Lamoreaux in U.S. Pat. No. 3,220,970, Karstedt in U.S. Pat. No. 3,814,730, and Modic in U.S. Pat. No. 3,516,946.

In an especially preferred embodiment, the addition curable silicone composition further includes a reinforcing organopolysiloxane resin of the type disclosed in U.S. Pat. No. 3,284,406

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to Nelson or U.S. Pat. No. 3,436,366 to Modic. Briefly, such resins are copolymers of SiO_2 units, $(\text{CH}_3)_3\text{SiO}_{0.5}$ units and $(\text{CH}_3)_2(\text{CH}_2=\text{CH})\text{SiO}_{0.5}$ units, and SiO_2 units, $(\text{CH}_3)_3\text{SiO}_{0.5}$ units and $(\text{CH}_3)(\text{CH}_2=\text{CH})\text{SiO}$ units, respectively. Particularly preferred organopolysiloxane resins are MDQ resins having vinyl unsaturation on monofunctional siloxane units, difunctional siloxane units, or both. The use of such reinforcing organopolysiloxane resins is especially desirable when the viscosity of the alkenyl containing polydiorganosiloxane is less than about 5000 centipoise at 25°C.

It is also contemplated that there may be included any conventional extending and/or reinforcing fillers. Fumed silica has been found to be particularly effective as a reinforcing filler for the silicone component of the present invention.

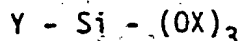
In another particularly preferred embodiment of the present invention, the addition curable silicone composition also contains a silane or polysiloxane which functions both as an inhibitor and as an adhesion promoter. One such composition is described in U.S. Pat. No. 3,759,968 to Berger et al. as a maleate or fumarate functional silane or polysiloxane. Compositions effective only as an inhibitor are disclosed in U.S. Pat. No. 4,256,870 to Eckberg and 4,061,609 to Bobear. Other suitable inhibitors will be obvious to those skilled in the art.

It is further contemplated that component (b) can be condensation curable silicone composition. Generally, condens-

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ation curable silicone compositions are available in either one or two packages and comprise (1) a polydiorganosiloxane having terminal hydrolyzable groups, e.g., hydroxyl or alkoxy, and (2) a catalyst which promotes condensation curing. Such compositions are well known in the art, for example, as described in U.S. Pat. No. 3,888,815 to Bessmer et al.

Alternatively, the polysiloxane network can be prepared by the hydrolytic polycondensation of silanes having the general formula



where each X is independently selected from the group consisting of hydrogen, alkyl radicals, hydroxyalkyl radicals, alkoxyalkyl radicals, and hydroxyalkoxyalkyl radicals, and Y is an alkyl radical, OX, where X is as previously defined, or an amino or substituted amino radical. The use of silanes having hydrolyzable groups to form a polysiloxane network of an interpenetrating polymer network is discussed in greater detail in U.S. Pat. No. 4,250,074 to Foscante et al.

The amount of curable silicone composition used in practicing the present invention can range from as little as about 1 part by weight per 100 parts by weight of component (a) to as much as about 150 parts by weight per 100 parts by weight of component (a). A particularly preferred embodiment of the present invention utilizes from 1 to 50 parts by weight of silicone per 100 parts by weight of component (a). Optimal results are obtained when from about 5 to about 35 parts by weight of silicone composition are used per 100 parts by weight of polytetrafluoroethylene. It should be noted that the translucency of the final product increases as the ratio of silicone composition to PTFE increases.

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For ease of discussion, preparation of compositions having interpenetrating matrices in cured form will be described only with respect to polytetrafluoroethylene and addition curable silicone compositions. Those skilled in the art will appreciate that in order to prevent premature curing of the silicone composition the components must not all be combined until the time of use unless a suitable inhibitor is included (or moisture excluded in the case of condensation curable silicone compositions).

In a preferred embodiment of this invention a one component, addition curable silicone composition containing an amount of inhibitor effective to prevent curing below about 100°C is dissolved in a suitable solvent, for example, kerosene or mineral spirits. The resulting solution is thereafter mixed with the polytetrafluoroethylene powder in a tumbler-type mixer suitable for mixing liquids with solids in order to incorporate the desired level of silicone into the PTFE. The semi-dry powder obtained is a mixture of PTFE, silicone composition and solvent which can be pressed into a cylindrical bar or other suitable shape.

The cylindrical bar is then extruded and calendered to provide a film of desired thickness. After the solvent is removed the film can be stretched to provide a film having a porous microstructure consisting of nodes and fibrils. Although expansion of the film can be carried out at temperatures ranging from room temperature to about 325°C, it is preferable that the temperature range from about 250°C to about 300°C so that the silicone composition cures during the stretching process.

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5 If an inhibitor is not present to prevent premature curing of the silicone composition, it is desirable to prepare a mixture of PTFE and vinyl containing polydiorganosiloxane and a mixture of PTFE and organohydrogenpolysiloxane, either or both mixtures also containing a curing catalyst. At the time of use the powders are combined and the final product prepared as if an inhibitor were present.

10 Because of the presence of the silicone in the extruded PTFE film, stretching was easier as the silicone acted as an internal plasticizer. Consequently, it is not required that stretching be effected as fast as possible, and excellent results have been obtained with stretch rates of only 20% per second. Of course, this does not preclude stretching at rates of 1000% per second or more as taught by Gore in U.S. Pat. No. 15 3,953,566.

20 Stretched films prepared in accordance with the present invention generally have thicknesses ranging from about 0.5 mils to about 10 mils. Quite surprisingly, however, the resultant materials had larger pore sizes than similar materials prepared solely from PTFE, yet they exhibit improved air permeability and improved resistance to liquid water permeability.

25 It is also preferred that the expanded products be heated to above the lowest crystalline melting point of the PTFE so as to increase the amorphous content of the polymer, typically to 10% or more. Such amorphous regions within the crystalline structure appear to greatly inhibit slippage along the crystalline axis and thereby lock fibrils and nodes so that they resist slippage under stress. As a result, a surprising 30 increase in strength is obtained.

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In another aspect of the present invention there are provided compositions having interpenetrating matrices in cured form comprising:

(a) a first polymer network comprising
5 polytetrafluoroethylene, and

(b) a second polymer network comprising
polydiorganosiloxane.

Components (a) and (b) of this aspect of the invention are the same as described hereinabove, however, the final product
10 is prepared by conventional extrusion techniques rather than by stretching.

Compositions prepared in accordance with the present invention find particular utility as filters, pump packing, insulation for electrical cables, and as laminates useful in
15 the manufacture of breathable wearing apparel.

In order to better enable those skilled in the art to practice the present invention, the following examples are provided by way of illustration and not by way of limitation. All parts and percentages are by weight unless otherwise noted.

20 EXAMPLES

Silicone compositions comprising the components set forth in Table 1 were prepared and identified as silicone A through silicone J.

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Table 1 - Silicone Compositions Evaluated as Additives to Polytetrafluoroethylene

Ingredients	Experimental Formulations Evaluated with Polytetrafluoroethylene (PTFE) (X Composition)									
	A	B	C	D	E	F	G	H	I	J
$\text{V1SiMe}_2\text{O}[\text{SiMe}_2\text{O}]_n\text{SiMe}_2\text{V1}$ (3,500 cps)	68.2	75	-	-	-	-	34.5	-	-	-
$\text{V1SiMe}_2\text{O}[\text{SiMe}_2\text{O}]_n\text{SiMe}_2\text{V1}$ (30,000 cps)	-	-	-	-	-	-	-	14.2	23.7	47.4
M'QD Resin (M' = Me ₂ HSiO, Q = SiO ₄)	22.7	25	-	-	-	-	36.4	-	-	-
M'Q Resin + M'D ₂ M' (M' = Me ₂ HSiO) (50:50)	8.2	-	100	-	-	-	8.2	0.64	1.1	2.2
Alc(trimethoxysilylpropyl)malate	0.9	-	-	-	-	-	0.9	0.15	0.25	0.5
$\text{Me}_3\text{SiO}[\text{SiMe}_2\text{O}]_n\text{SiMe}_3$ (1,000 cps)	-	-	-	100	-	-	-	-	-	-
$\text{Me}_3\text{SiMe}_2\text{O}[\text{SiMe}_2\text{O}]_n\text{SiMe}_2\text{OH}$ (3,000 cps)	-	-	-	-	100	-	-	-	-	-
$\text{Me}_3\text{Si}(\text{OMe})_2\text{O}[\text{SiMe}_2\text{O}]_n\text{Si}(\text{OMe})_2\text{Me}$ (3,500 cps)	-	-	-	-	-	100	-	-	-	-
Toluene	-	-	-	-	-	-	-	85	75	50
Platinum Catalyst (Lamorex)	10 ppm	10 ppm	-	-	-	-	10 ppm	1.0 ppm	1.0 ppm	1.0 ppm

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Silicone - PTFE compositions were then prepared by dissolving the silicone in kerosene and then mixing this solution with PTFE in a tumbler-type mixer. These compositions are set forth in Table 2 as compositions 1 through 10.

5

Table 2 - Silicone - PTFE Compositions

Composition Number	% Kerosene ¹	% Silicone ²	Weight Kerosene, lbs.	Weight Silicone, lbs.	Weight PTFE, lbs.	Weight CaCO ₃ , lbs.
1	17	7	11	4.5	60	-
2	17	7	1.2	0.5	6.5	-
10 3	17	7	1.2	0.5	6.5	-
4	17	7	1.2	0.5	6.5	-
5	17	7	1.2	0.5	6.5	-
6	17	7	1.2	0.5	6.5	-
7	17	20	1.2	1.4	5.6	-
15 8	17	7	1.2	0.5	6.2	0.35
9	17	7	1.2	0.5	6.5	-
10 10	13	20	1.2	1.4	5.6	-

1. Kerosene calculated as percent of total solids weight.
2. Silicone level based on percent of final "dried", extruded film.

20

Extrusions of the foregoing silicone - PTFE compositions were prepared as specified in Table 3.

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Table 3 - PTFE - Silicone Extrudates

	<u>Extruded Sample No.</u>	<u>Silicone Used (%)</u>	<u>Stretch Ratio</u>	<u>Specific² Gravity</u>	<u>Final Width, ins.</u>
5	1A	A(7)	1:1	1.63	6.4
	1B	A (7)	1:2	1.09	5.9
	1C	A (7)	1:3	0.87	5.4
	1D	A (7)	1:4	0.77	4.9
	2	D (7)	1:1	1.23	5.5
10	3	E (7)	1:1	1.29	6.0
	4	F (7)	1:1	1.38	5.75
	5	B (7)	1:1	1.32	6.0
	6A	E+C(7)	1:1	-	5.75
	6B	B+C (7)	1:3	1.41	5.75
15	7	A (20)	1:1	1.75	4.0
	8	A (7, 5% CaCO ₃)	1:1	1.70	6.0
	9	G(7)	1:1	1.66	6.75
	10 ³	B+C (7)	1:1	1.95	6.0
	11	G (20)	1:1	-	-

20

1. All samples used 17% kerosene as a lubricant except 11 which used 13% resulting in a more difficult extrusion.

2. The reduction in specific gravity for 1A - 1D reflects the difference in longitudinal stretching of the samples.

25

3. Extrudable PTFE-Silicone mixture stored at 72°F for 22 hours before extrusion.

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5 The extruded, calendered and dried PTFE - silicone samples were then stretched from their extruded width of about six inches to 64, 66 or 74 inches. The films were sintered during the stretching operation to prevent shrinking. Moreover, since sintering was effected at 300°C, the crosslinkable silicone mixtures were crosslinked during this operation. Physical properties of the stretched film are shown in Table 4 where data on permeability of the stretched films with different silicone additives are compared.

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Table 4 - Physical Properties and Permeability Data for Stretched PIPE-Silicone and PTFE Film

Sample Number	Stretch Ratio	Initial Width, in.	Final Width, in.	Film Thickness, mil	Specific Gravity	Mullen Value, psi	Bubble Point, psi	Maximum Pore Size, microns	Air Permeability Cu Ft/m ² /in. ² /24 hr	Moisture Vapor Transmission g/m ² /24 hr
1A	1:1	4.5	74	1.4	0.49	26	4.0	0.809	1.0	794
1A	1:1	4.5	66	0.8	0.57	18	2.0	1.62	3.0	873
1B	1:2	4.0	64	0.3	1.02	49	8.0	0.304	2.0	538
1B	1:2	4.0	74	0.4	0.73	39	7.0	0.462	2.0	831
1C	1:3	3.5	74	0.2	0.96	93	12.0	0.270	3.0	843
1D	1:4	3.5	74	0.2	0.96	65	11.0	0.294	2.5	816
2	1:1	3.5	74	0.4	0.69	26	5.0	0.847	4.0	867
3	1:1	4.0	74	1.0	0.33	23	3.0	1.08	3.5	867
4	1:1	4.25	64	1.3	0.30	25	5.0	0.647	2.0	642
5	1:1	4.0	64	2.1	0.36	23	5.0	0.647	1.5	890
5	1:1	4.0	74	1.0	0.33	33	5.0	0.647	2.0	830
6A	1:1	4.25	64	-	-	-	-	-	-	-
6B	1:3	4.25	64	-	-	-	-	-	-	-
7	1:1	3.25	64	-	-	-	-	-	-	-
8	1:1	4.0	64	-	-	-	-	-	-	-
9	1:1	4.5	64	2.6	0.35	16	3.5	0.901	2.0	743
PTFE	1:1	4.5	74	1.5	0.4	35	7.0	0.50	1.5	900
PTFE	1:2	4.5	74	1.5	0.35	20	13.0	0.20	1.0	900

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In Table 4 the stretched film property measurements have the following meanings:

- 5 Stretch Ratio - Since the extrudate is conveyed over heated drum-rollers during the drying step, longitudinally stretching of the film can be varied by adjusting the speed of the drums. A ratio of 1:1 assumes that no stretching occurs in the machine direction.
- 10 Mullen Value - Measures the pounds per square inch of water pressure needed to get penetration of the film. A measure of waterproofness. The higher the Mullen value the more waterproof is the film. A minimum value of 25 is desired.
- 15 Bubble Point - A measure of the pounds per square inch of pressure necessary to produce a bubble in a column of water on the opposite side of the film. This is a measure of the porosity from which the maximum pore size can be calculated.
- 20 Air Permeability - Measures the breatheability of the film by the air flow through the membrane expressed in cubic feet per minute per square foot of area.
- 20 Moisture-Vapor Transmission (MVT) - This also measures the breatheability of the film by the number of grams of water evaporated per square meter of area over a 24-hour period when air flows across a film in contact with water on the opposite side. The higher the value the greater the breatheability.

25 The results in Table 4 show that excellent results are obtained in comparison to stretched PTFE which does not have an interpenetrating polydiorganosiloxane network.

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5 A detailed set of extraction experiments were then conducted to show that the silicone compositions did cure into the PTFE matrix. Also included in the extraction study were silicones which could not be crosslinked. Thus, weight loss studies before and after extraction with hexane in a Soxhlet extraction apparatus show whether an interpenetrating network was formed. The data were collected for both extruded and stretched samples and the results as set forth in Table 5.

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Table 5 - Extraction and Elemental Analysis Results for Extruded and Stretched PTFE-Silicone Compositions

Sample Number	Type of PTFE-Silicone Film	Heat Treatment	X Silicone (Theory)	X Silicone (Extracted)	X Si (Theory)	X Si (Calculated)	X Si (Found)
1A	Extruded	None	7.0	1.01	2.65	2.27	2.37
1A	Extruded	200°C, 1 hr	7.0	0.630	2.65	2.41	2.65
1A	Stretched	Sintered, ~300°C	7.0	0.523	2.65	2.45	2.83
7	Extruded	None	20.0	1.34	7.57	7.06	6.20
7	Extruded	None	20.0	2.275	7.57	6.71	6.42
7	Extruded	200°C, 1 hr	20.0	0.41	7.57	7.41	6.72
8	Extruded	None	7.0	1.375	2.65	2.13	2.09
8	Extruded	200°C, 1 hr	7.0	7.01	2.65	0	0.51
8A	Extruded	None	7.0	0.78	2.65	2.35	2.58
9	Extruded	None	7.0	0.736	2.65	2.37	2.47
9	Extruded	200°C, 1 hr	7.0	0.90	2.65	2.31	3.73
2	Extruded	None	7.0	6.97	2.65	0.011	0.35
3	Extruded	None	7.0	7.03	2.65	0	0.34
3	Extruded	Sintered, ~300°C	7.0	6.23	2.65	0.257	1.37
4	Extruded	None	7.0	2.77	2.65	1.60	1.72

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In Table 5, the theoretical weight percent silicone is that amount mixed with PTFE, while the percent silicone extracted is the actual weight loss from the sample after extraction. From these two values percent silicon for the amount originally added (% Si theory) and percent silicon for the amount remaining after extraction (% Si calculated) were determined. These results were then compared with the percent silicon found by elemental analysis (% Si found). For sample 5, which contained an uncrosslinkable silicone, essentially all of the silicone was extracted, even after thermal treatment.

A comparison of the stretched PTFE - silicone films shows that for the 1A sample essentially none of the silicone could be extracted, whereas for sample 3 most of the silicone was extracted as expected.

Sample 4 showed less extractable silicone than was expected, suggesting that crosslinking occurred through interaction of the material with atmospheric moisture or moisture introduced during processing, since this was a condensation curable silicone.

Next, an extruded sample of PTFE having a 50 mil thickness and containing 20 percent of Silicone G (Table 1) was cured at 200°C for two hours. The resultant material had a Shore A hardness of 90, 484 psi tensile at break and 183% elongation. In addition to being elastomeric, the material was substantially translucent.

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I Claim:

1. A composition having interpenetrating matrices, comprising:

(a) a first polymer network characterized by nodes interconnected by fibrils, and

(b) a second polymer network comprising diorganosiloxane units.

2. A composition as in Claim 1, wherein component (a) is selected from the group consisting of polytetrafluoroethylene, polyethylene, polyamides and polyesters.

3. A composition as in Claim 1, wherein component (a) is polytetrafluoroethylene.

4. A composition as in Claims 1 or 3 wherein component (b) is formed from an addition curable silicone composition comprising (1) a polydiorganosiloxane having alkenyl unsaturation, (2) an organohydrogenpolysiloxane crosslinking agent, and (3) a catalyst for promoting crosslinking of (1) and (2).

5. A composition as in Claim 4, wherein the viscosity of the polydiorganosiloxane having alkenyl unsaturation is less than 100,000 centipoise at 25°C.

6. A composition as in Claim 4, wherein the viscosity of the polydiorganosiloxane having alkenyl unsaturation is from about 500 centipoise to about 50,000 centipoise at 25°C.

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7. A composition as in Claim 4, wherein the viscosity of the polydiorganosiloxane having alkenyl unsaturation is from about 3000 centipoise to about 6000 centipoise at 25°C.

8. A composition as in Claim 5, wherein the organohydrogenpolysiloxane has a viscosity of from about 25 centipoise to about 10,000 centipoise at 25°C.

9. A composition as in Claim 6, wherein the organohydrogenpolysiloxane has a viscosity of from about 100 centipoise to about 1000 centipoise at 25°C.

10. A composition as in Claim 9, wherein the organohydrogenpolysiloxane is substantially linear.

11. A composition as in Claim 9, wherein the organohydrogenpolysiloxane is a resin.

12. A composition as in Claim 8, wherein the curing catalyst is a platinum containing complex.

13. A composition as in Claim 12, further comprising a reinforcing organopolysiloxane resin.

14. A composition as in Claim 13, wherein the reinforcing organopolysiloxane resin has vinyl unsaturation on mono-functional siloxane units, difunctional siloxane units, or both.

15. A composition as in Claim 8, further comprising a filler.

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16. A composition as in Claim 15, wherein the filler is fumed silica.

17. A composition as in Claim 8, further comprising a cure inhibitor.

18. A composition as in Claim 17, wherein the cure inhibitor is a maleate or fumarate functional silane or polysiloxane.

19. A composition as in Claims 1 or 3, wherein component (b) is formed from a condensation curable silicone composition.

20. A composition as in Claims 1 or 3, wherein the weight ratio of component (b) to component (a) is from 1:100 to 3:2.

21. A composition as in Claims 1 or 3, wherein the weight ratio of component (b) to component (a) is from 1:100 to 1:2.

22. A composition as in Claims 1 or 3, wherein the weight ratio of component (b) to component (a) is from 1:20 to 7:20.

23. A composition having interpenetrating matrices, comprising:

(a) a first polymer network comprising polytetrafluoroethylene; and

(b) a second polymer network comprising polydiorganosiloxane.

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24. A composition as in Claim 23, wherein the polydiorganosiloxane is formed from an addition curable silicone composition.

25. A composition as in Claim 23, wherein the polydiorganosiloxane is formed from a condensation curable silicone composition.

26. A composition as in Claim 23, wherein the weight ratio of component (b) to component (a) is from 1:100 to 3:2.

27. A composition as in Claim 23, wherein the weight ratio of component (b) to component (a) is from 1:100 to 1:2.

28. A composition as in Claim 23, wherein the weight ratio of component (b) to component (a) is from 1:20 to 7:20.

29. A composition capable of being processed to a material having interpenetrating matrices, comprising a mixture of

- (a) polytetrafluoroethylene, and
- (b) a curable silicone composition.

30. A composition as in Claim 29, further comprising a solvent.

31. A composition as in Claim 29, wherein component (b) comprises (1) a polydiorganosiloxane having alkenyl unsaturation, (2) an organohydrogenpolysiloxane crosslinking agent, and (3) a catalyst for promoting crosslinking of (1) and (2).

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32. A composition as in Claim 31, wherein the viscosity of the polydiorganosiloxane having alkenyl unsaturation is less than 100,000,000 centipoise at 25°C.

33. A composition as in Claim 31, wherein the viscosity of the polydiorganosiloxane having alkenyl unsaturation is from about 500 centipoise to about 50,000 centipoise at 25°C.

34. A composition as in Claim 31, wherein the viscosity of the polydiorganosiloxane having alkenyl unsaturation is from about 3000 centipoise to about 6000 centipoise at 25°C.

35. A composition as in Claim 33, wherein the organohydrogenpolysiloxane has a viscosity from about 100 centipoise to about 1000 centipoise at 25°C.

36. A composition as in Claim 35, wherein the organohydrogenpolysiloxane is a resin.

37. A composition as in Claim 31, wherein the catalyst is a platinum containing complex.

38. A composition as in Claim 29, further comprising a reinforcing organopolysiloxane resin.

39. A composition as in Claim 29 or 38, further comprising a filler.

40. A composition as in Claim 29 or 38, further comprising a cure inhibitor.

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41. A composition as in Claim 29, wherein the curable silicone composition is condensation curable.

42. A composition as in Claim 29, wherein the weight ratio of component (b) to component (a) is from 1:100 to 3:2.

43. A composition as in Claim 29, wherein the weight ratio of component (b) to component (a) is from 1:100 to 1:2.

44. A composition as in Claim 29, wherein the weight ratio of component (b) to component (a) is from 1:20 to 7:20.

45. An article of manufacture made from the composition of Claims 1, 3, 23 or 29.

46. A method for making a composition capable of being processed into a material having interpenetrating matrices in cured form, comprising the steps:

(a) dissolving a curable silicone composition in a solvent, and

(b) mixing the solution obtained in (a) with polytetrafluoroethylene powder.

47. A method as in Claim 46, wherein the solvent is kerosene, toluene, or mineral spirits.

48. A method as in Claim 46, wherein mixing is effected in a tumbler-type mixer suitable for mixing liquids with solids.

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49. A method for making a composition having interpenetrating matrices, comprising the steps:

- (a) expanding a shaped article comprising a polymer capable of being expanded to a material characterized by nodes interconnected by fibrils and a curable silicone composition and
- (b) curing said curable silicone composition.

50. A method as in Claim 49, wherein steps (a) and (b) are carried out simultaneously.

51. A method as in Claims 49 or 50 wherein the polymer capable of being expanded to a material characterized by nodes interconnected by fibrils is selected from the group consisting of polytetrafluoroethylene, polyethylene, polyamides and polyesters.

52. A method as in Claims 49 or 50, wherein the polymer capable of being expanded to a material characterized by nodes interconnected by fibrils is polytetrafluoroethylene.

53. A method as in Claims 49 or 50 wherein the curable silicone composition is an addition curable silicone composition.

54. A method as in Claim 53, wherein curing is effected at a temperature above 100°C.

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55. A method as in Claim 54, wherein curing is effected at a temperature of from about 250°C to about 300°C.

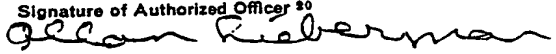
56. A method as in Claim 52, further comprising the step of heating the expanded material to above the lowest crystalline melting point of the polytetrafluoroethylene.

57. A method as in Claim 49, wherein step (a) is carried out by expanding the shaped article at a stretch rate of about 20% per second.

58. A method as in Claim 49, wherein step (a) is carried out by expanding the shaped article at a stretch rate of at least 1000% per second.

INTERNATIONAL SEARCH REPORT

International Application No PCT/US86/02405

I. CLASSIFICATION OF SUBJECT MATTER (If several classification symbols apply, indicate all) ³		
According to International Patent Classification (IPC) or to both National Classification and IPC		
IPC(4): C08K 3/36, 5/01; C08L 27/18, 83/06		
US CL. 264/127; 524/506; 525/104, 903.		
II. FIELDS SEARCHED		
Minimum Documentation Searched ⁴		
Classification System	Classification Symbols	
U.S.	525/104, 903; 524/506; 264/127	
Documentation Searched other than Minimum Documentation to the Extent that such Documents are Included in the Fields Searched ⁵		
III. DOCUMENTS CONSIDERED TO BE RELEVANT ¹⁴		
Category [*]	Citation of Document, ¹⁶ with indication, where appropriate, of the relevant passages ¹⁷	Relevant to Claim No. ¹⁸
X	US,A, 2,710,290 (SAFFORD) 07 JUNE 1955, See column 1, lines 51-66; column 2, lines 1-12; column 5, lines 35-44 and 55-67; column 6, lines 1-19.	1-45,49-58
X	US,A, 2,927,908 (KONKLE) 08 MARCH 1960, See column 1, lines 15-18, column 2, lines 69-72; column 3, lines 1-36.	1-45,49-58
X	US,A, 2,934,515 (KONKLE) 26 APRIL 1960, See column 1, lines 56-70; column 3, lines 21-40.	1-11,20-29,31-36 42-45,49-58
X	US,A, 3,207,825 (TULLY) 21 SEPTEMBER 1965, See column 2, lines 6-25 and 47-60.	1-58
A	US,A, 3,865,897 (FALENDER) 11 FEBRUARY 1975, See column 2, lines 28-44; column 5, lines 50-55; column 6, lines 8-14.	45,49-51,53-55 57, 58
Y	US,A, 3,962,153 (GORE) 08 JUNE 1976, See column 1, lines 34-43; column 2, lines 9-18 and 53-55.	1-58
Y	US,A, 4,010,136 (BLIZZARD) 01 MARCH 1977, See column 1, lines 63-68; column 2, lines 1-3; column 5, lines 12-50.	1-58
<p>[*] Special categories of cited documents: ¹⁵</p> <p>"A" document defining the general state of the art which is not considered to be of particular relevance</p> <p>"E" earlier document but published on or after the international filing date</p> <p>"L" document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)</p> <p>"O" document referring to an oral disclosure, use, exhibition or other means</p> <p>"P" document published prior to the international filing date but later than the priority date claimed</p> <p>"T" later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention</p> <p>"X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step</p> <p>"Y" document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art.</p> <p>"&" document member of the same patent family</p>		
IV. CERTIFICATION		
Date of the Actual Completion of the International Search ¹	Date of Mailing of this International Search Report ²	
28 JANUARY 1987	09 FEB 1987	
International Searching Authority ¹	Signature of Authorized Officer ²⁰	
TSA/US	 ALLAN LIEBERMAN	

Form PCT/ISA/210 (second sheet) (October 1981)

III. DOCUMENTS CONSIDERED TO BE RELEVANT (CONTINUED FROM THE SECOND SHEET)		
Category *	Citation of Document, ¹⁶ with indication, where appropriate, of the relevant passages ¹⁷	Relevant to Claim No ¹⁸
Y	US,A, 4,096,227 (GORE) 20 JUNE 1978, See column 1, lines 35-50; column 2, lines 53-56.	1-58
Y	US,A, 4,187,390 (GORE) 05 FEBRUARY 1980, See column 1, lines 40-45; column 3, lines 1-15 and 34-37.	1-58
A	US,A, 4,336,364 (MAXSON) 22 JANUARY 1982, See column 3, lines 40-64.	13,17,31,37 40, 41
Y	US,A, 4,347,204 (TAKAGI) 31 AUGUST 1982, See column 2, lines 56-68; column 3, lines 6-17.	1-58
X	US,A, 4,478,665 (HUBIS) 23 OCTOBER 1984, See column 2, lines 21-38; column 3, lines 6-12.	1-45,49-58
A	US,A, 4,544,692 (KUZIEMKA) 01 OCTOBER 1985. See the entire document.	1-58